# MORPHOLOGICAL AND FLEXURAL PROPERTIES OF LIGHTWEIGHT GYPSUM BASED FIBER REINFORCED COMPOSITE 

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#### Abstract

This paper describes the results of the experimental investigations of the properties of lightweight gypsum based fiber reinforeed composites. Two gencral types of fiber reinforcement were examined, namely, polymer based fibers and natural fibers. The study is essentially a developmental research with the objective of evaluating the properties of natural fibers in comparison with synthetic fibers of the same density in a gypsum matrix. A gypsum-cellulose pulp and a gypsum-Shirasu balloon sand were used as lightweight gypsum matrices. Experimental and analytical results on both standard specimens and full scale structural elements such as wall panel are presented.


## INTRODUCTION

Fiber reinforced composite or fiber concrete is essentially a hydraulic cement material (cement or gypsum binder) with aggregates and in the main discrete fiber reinforcement. The concept of addition of fibers into a brittle matrix is an attempt: (a) to locally restrain the propagation of internal cracks or flaws into adjacent material, (b) to delay and control tensile cracking, (c) to dissipate energy at its interfacial region, and (d) to improve the thixotropic and rheological properties of plain matrix (RILEM, 1990; Hannant, 1981; Swamy, 1990). However, the degree and the effectiveness of reinforcement depends on the type of fiber. Generally, fiber suitable of reinforcement are divided into categories of high modulus and low modulus fibers. High modulus fibers such as glass or steel are capable of producing strong composites with high stiffness and load carrying capacity. Low modulus fibers such as polypropylene and natural fibers on the other hand, are capable of producing composites with high ductility and toughness with little increase in strength.

The relatively high cost of man-made or synthetic fibers such as polypropylene, polyethylene, and glass used in fiber reinforced composites make it desirable to evaluate natural fibers such as sisal and coconut fibers as possible substitutes. The author believes that the
potential of natural fiber composite in building construction is very significant. Fiber cement sheeting, panel and precast building components are few of many possible applications of natural fiber cement composites. However, at present there are very limited development on the utilization of this type of composites especially with $\alpha$-gypsum cement as binder mainly due to scarcity in information on the precise structure-properties of the composite system, especially in the form of structural elements.

Gypsum is obtained by calcining raw gypsum $\left(\mathrm{CaSO}_{4}-2 \mathrm{H}_{2} \mathrm{O}\right)$ at around $175^{\circ} \mathrm{C}$, and if water is added the resulting hemihydrate ( $\alpha$-gypsum or $\beta$-gypsum) rehydrates back to the dehydrate form in about 20-30 minutes (Joshi et al., 1992; Singh et al., 1992, Evans et al., 1980). Currently, gypsum cement is categorized as either $\alpha$-gypsum or $\beta$-gypsum depending on the product of processing. $B$-hemihydrate gypsum is more commonly used in building construction in the form of pre-cast elements because of its slower setting time rate and lower cost compared to $\alpha$-gypsum, but it is relatively of lower strength. In this study, the feasibility of using $\alpha$-hemihydrate gypsum as the binding medium for building elements was investigated. The lightweight and fire resisting property of gypsum cement plaster makes it very suitable for panel elements in building construction. The low alkalinity of gypsum compared to Portland cement makes it an advantageous cementing material for natural fiber cement composites, where alkali attack on cellulose components of fibers is most critical.

Hence, a developmental and comparative study was undertaken to examine the properties of a lightweight gypsum based composite reinforced by polymer based fibers (vinylon and polypropylene), and natural fibers (sisal and coconut). The study is directed towards the investigation of properties of a fiber composite in the form of structural pre-cast element such as wall panel. This experiment was conducted with the objective to:
(a) examine the morphological properties of gypsum matrix and the aggregate constituents,
(b) examine the effect of fiber type to the mechanical properties of gypsum based matrix,
(c) examine the effect of lightweight aggregates and mineral admixtures to the properties of fiber composite.

## EXPERIMENTAL PROGRAM

In order to limit the experimental program to more significant variables, previous work done by the author on natural fiber concrete were considered (Jorillo, 1993; Shimizu and Jorillo, 1992). In this study the variables considered in the testing program are listed in Table 1.

Physical property tests like x-rays, fluorescence spectral analysis and scanning electron microanalysis were conducted. Likewise, mechanical property tests such as compression, bending, and deformation analysis were also undertaken in accordance with ASTM standards.

The following information obtained from the tests are:
(a) Fiber and interface morphological analysis
(b) Stress-deformation diagram considering the load at first crack and post-cracking stage and,
(c) Mechanical properties of composite.

## Materials and Testing Methods

$\alpha$-gypsum ( $\alpha-\mathrm{CaSO}_{4}+1 / 2 \mathrm{H}_{2} \mathrm{O}$ ) with $95 \%$ concentration (ASTM C-22) was used as binding medium. The chemical properties are shown in Table 2. Processed cellulose pulp from recycled paper and sankelite sand (Shirasu balloon of volcanic origin) with dry rodded density of 90 and $80 \mathrm{~kg} / \mathrm{cm}^{3}$ respectively were used as lightweight fine aggregates. An inorganic retarder with chemical composition of $\mathrm{P}_{2} \mathrm{O}_{5}, \mathrm{SO}_{3}$ and CaO of $1.7,59.8$ and $38.5 \%$ respectively was used at a dosage rate of $1 \%$ by weight of gypsum. Mineral admixture such as silica fume ( S ), white silica (CS), and fly-ash (F) were used with gypsum-pulp mortar. The chemical properties of these materials as determined from X-ray fluorescence test are enumerated in Table 3. Also, typical x-ray fluorescence results for the chemical analysis of silica fume and white silica are shown in Figures 2(a) and 2(b) respectively., Furthermore, SEM micrograph of the materials (e.g. fly-ash, white silica, cellulose pulp, and sankelite) are shown in Figures 3-6.

Table 1
Variables Considered in the Gypsum Composite Test Program

| Parameter | Variable | Range | Limitations* |
| :---: | :---: | :---: | :---: |
| Gypsum Binder (G) | Water/Gypsum (W/G) | $\mathrm{W} / \mathrm{G}=35,45 \%$ | $\alpha$-Gypsum cement |
| Aggregates (Sand/Binderratio) | Gypsum-Cellulose (GP) <br> Gypsum-Sankelite (GS) | Cellulose Pulp $P=10,15,20,30 \%$ | Lightweight Pulp GP- series Lightweight Sand GS-series |
| Mineral Admixture (\% substitution by wt. of cement) | Silica Fume (S) <br> White Silica (CS) <br> Fly-ash (F) | $\begin{aligned} & S /(S+G)=5,10 \% \\ & C S /(C S+G)=5,10 \% \\ & F /(F+G)=5,10 \% \end{aligned}$ | Tests conducted with mineral admixture are for GP Mix series with Pulp sand only |
| Fiber Reinforcement <br> A. Type of Fiber <br> B. Fiber Proportion | 1. Polymer-based fibers <br> 2. Natural fibers <br> $\mathrm{Vf}=(\%$ volume of matrix $)$ | 1. Vinylon, Polypropylene <br> 2. Coconut, Sisal $\text { If }=8-12 \mathrm{~mm}$ <br> $\mathrm{Vf}=1,2 \%$ $\mathrm{Vf}=2,3 \%$ | Mix series with Pulp Mix series with Sankelite <br> Mix series with Pulp Mix series with Sankelite |
| Specimen Preparation | Ordinary casting method | Specimen dimension $\begin{aligned} & 40 \times 40 \times 160 \\ & 20 \times 150 \times 300 \\ & 20 \times 300 \times 600 \end{aligned}$ | All Mix series <br> Mix series with Sankelite |

[^0]Table 2
Physical and Chemical Properties of $\alpha$-Gypsum

| Properties |  | Ave. Values | Oxides** | Value (\%) |
| :---: | :---: | :---: | :---: | :---: |
| Specific Gravity |  | 2.40 | $\mathrm{SO}_{3}$ | 60.1 |
| Blaine's <br> Fineness ( $\mathrm{cm}^{2} / \mathrm{g}$ ) |  | 3000 | CaO | 34.7 |
| Consistency * <br> and <br> Scting <br> Time | Stucco-water consistency (\%) | 40 | $\mathrm{Re}_{2} \mathrm{O}$ | 5.1 |
|  | Init setting time (h-min) | 0-14 |  |  |
|  | Final scting time (h-min) | 0-20 |  |  |
| Strength ( $\mathrm{kg} / \mathrm{cm}^{2}$ ) | Bending strength | 23.0 |  |  |
|  | Compressive strength | 240 |  |  |

* per ASTM C-22 specification
* Chemical composition listed are summarized from the $x$-ray fluorescence spectral analysis and not from the oxides analysis prescribed by ASTM

Table 3
Chemical Composition of Mineral Admixture and Aggregate

| Chemical compound | Silica Fume | White Silica | FLy-ash | Coarse Cellulose Pulp | Finc <br> Cellulose Pulp | Sankelite <br> Shirasu sand |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| Na 2 O | 0.15 | 0.12 | 0.10 | - | 1.8 | 0.5-2.0 |
| MgO | 0.29 | 0.22 | 0.30 | 2.4 | - | 0.4-0.6 |
| SiO 2 | 83.3 | 74.5 | 81.1 | 79.9 | 39.8 | 54-75 |
| CiO 2 | 0.21 | - | - | - | - | - |
| K2O | 15.8 | 16.8 | 0.60 | 15.7 | 17.1 | 0.1-4.0 |
| CaO | 0.071 | 0.18 | 1.40 | 0.91 | 0.40 | 1.5-2.5 |
| In2) | 0.22 | 0.17 | - | 0.18 | 0.19 | - |
| TiO 2 | - | 0.12 | - | - | 0.33 | - |
| Sb2O5 | - | 7.9 | - | - | 11.6 | - |
| Fe 2 O 3 | - | - | 1.0 | 0.44 | 1.2 | 1.0-4.0 |
| Al2O3 | - | - | 15.5 | - | 2.27 | 11-16 |
| P2O5 | - | - |  | - | - | - |

* Chemical composition listed are summarized from the $x$-ray fluorescence spectral analysis and not from the oxides analysis prescribed by ASTM

For fiber reinforcement, two general types of low density fibers were used, namely, polymer based fibers with specific gravity of 0.98-1.3 (i.e., vinylon and polypropylene) and natural fibers with specific gravity of 1.0-1.15 (i.e., coconut and sisal). The properties of these fibers are listed in Table 4. Coconut (cocos nucifera) fibers extracted from mature coconut fruit, and commercial grade sisal (agave sisalana) fibers were used for natural type of fibers. While, a hydrocarbon polymer based polypropylene fiber and polyvinyl alcohol derivative vinylon fibers were used for synthetic fiber reinforcement. SEM micrographs of fibers shown in Figures 7-10 reveal their morphological structure which governs the properties of the composite as a whole system.

Table 4
Properties of Fibers

| Fiber | Diameter <br> $(\mathrm{mm})$ | Length <br> $(\mathrm{mm})$ | Density <br> $\left(\mathrm{kg} / \mathrm{cm}^{3}\right)$ | Young's <br> Modulus <br> $(\mathrm{Gpa})$ | Tensile <br> Strength <br> $(\mathrm{Mpa})$ |  |
| :--- | :--- | :--- | :---: | :---: | :---: | :---: |
| Vinylon (PVA) <br> monofilament | $0.20-0.25$ | 12 | 1300 | 31 | Elongation <br> at Break <br> $(\%)$ |  |
| Polypropylene <br> strand type | $0.10-0.15$ | 8 | 910 | 12 | 500 | $6-8$ |
| Glass* <br> strand type | $0.15(30 \times)$ | 10 | 2700 | 70 | 1250 | $3-5$ |
| Coconut <br> monofilament | $0.20-0.45$ | 12 | 1180 | $8-12$ | $100-150$ | $20-25$ |
| Sisal <br> monofilament | $0.05-0.20$ | 12 | 1270 | $12-15$ | $400-460$ | $13-15$ |

* High modulus fiber used for comparison of strength properties


## Mix Proportioning, Method of Mixing and Casting

The adopted mix proportion has a $\mathrm{W} / \mathrm{C}=35$ and $45 \%$, sand-cement ratio ( $\mathrm{S} / \mathrm{C}$ ) of 10 , 15,20 , and $30 \%$, retarder admixture dosage of $1.0 \%$ by weight of gypsum, mineral admixture dosage of 5 and $10 \%$ by weight of cement, and a fiber volume of $1-4 \%$ per total volume of matrix. A simple mix code designation is adopted in the experiments, for example a code G35P10 means a plain gypsum mortar with $\mathrm{W} / \mathrm{G}=35 \%$ and pulp ( P ) content of $10 \%$. Likewise for fiber composite series a code V2-G45 refers to a vinylon fiber composite (V) with $2 \%$ fiber volume in a gypsum-pulp (G) with $\mathrm{W} / \mathrm{G}=45 \%$. For the series of gypsum-sankelite matrix a code of GS was used.

The concrete were mixed in a 5 -liter Hobart mixer in the following order, that is (1) half of the dry materials and three-fourths water, (2) gradual introduction for fibers.(half volume) and, (3) addition of remaining water and fiber. The total mixing time was about 10 minutes on the average. The designed workability of the base mortar is $180-200 \mathrm{~mm}$, which is workable enough to dispersed a fiber volume of up to $3 \%$. The mixture were then cast in its respective mold and stored in temperature controlled room ( $20^{\circ} \mathrm{C}, 80 \% \mathrm{RH}$ ) for 24 hours. After which the specimens were demolded and stored in $20^{\circ} \mathrm{C}$ water until the testing age of 7 and 28 days.

## Test Set-up

Bending and compressive properties were taken from $40 \times 40 \times 160 \mathrm{~mm}$ standard beam specimen and fractured beam specimen respectively in accordance with ASTM standards (ASTM C-348, C-349). As for the $20 \times 150 \times 300$ panel specimen, a special jig was made similar to the JIS A-1408 specification, as shown in Figure 1. Also, for larger panel specimen ( $20 \times 300 \times 600 \mathrm{~mm}$ ) four-point loading test set-up was used. To measure the modulus of elasticity and the Poisson's ratio of various specimen, a 30 mm strain gage were placed at the mid span of both compression and tensile side. Likewise, a pair of 25 mm (LVDT) transducers measured the mid span deformation of the specimen.



Load Point Detail Support Point Detail
(a) Test set-up for the $20 \times 150 \times 300 \mathrm{~mm}$ panel

(b) Test set-up for the $20 \times 300 \times 600 \mathrm{~mm}$ panel specimen

Figure 1
Flexural Set-up Adopted in the Experimental Program

## RESULTS AND DISCUSSION

## Morphological Properties of Gypsum-Pulp and Gypsum-Sankelite Fiber Composite

Microstructural study can provide plenty of insights and understanding to the structureproperty behavior of the composite system up to the engineering level. Through microstructural analysis, knowledge and information on the mechanism of the reinforcing action of the fiber, the transfer of stresses from and to the fiber, and the rheological relationships between aggregates, fibers and binder can be evaluated.

Figures 3 and 4 show the typical structure of fly-ash and white silica from Chuo Electric Plant, Japan (hereafter called white silica), respectively. Note the angular and porous structure, as well as the relatively bigger particle size of white silica compared to the smooth and round structure of fly-ash under the same SEM magnification. This morphology presumably may be one of the major reasons for the high water absorption, decreased workability of the paste, and relatively less contribution to strength development of mortar by the white silica compared to flyash.

Figures 5 and 6 show the typical structure of cellulose pulp processed from recycled paper and Shirasu balloon (Sankelite) lightweight sand of volcanic origin. It can be seen that the smooth and hollow structure of Shirasu sand are its major attributes in providing lightweightness and added workability to the matrix. The pulp sand on the other hand showed a highly porous and angular structure which consequently lead to high water absorption and reduced workability of the gypsum mixture. This appears to be the major reason why the gypsum-pulp matrix can only accept a maximum fiber volume addition of $2 \%$ compared to the $3 \%$ by the gypsumsankelite matrix at the same W/G and S/G ratio.

SEM micrographs in Figures 7-10 show the surface and section properties of the fiber reinforcement used in this study. It can be seen that sisal and coconut fibers basically possessed a multi-cellular structure typical of natural cellulose fibers. A single strand of fiber is composed of elongated oriented fibrils (principally crystalline cellulose) bound by a matrix (non-crystalline lignin complex). Both sisal and coconut fiber showed a fibril diameter range of $8-15 \mu \mathrm{~m}$. On the other hand, SEM micrographs (Figure 9) of polyvinyl alcohol (PVA) vinylon fiber showed a monofilament structure with fairly smooth and close surface with elongated marks left by polymer film drawing. Whereas, microstructure of the polypropylene fiber shows a synthetic multi-strand structure for additional bond strength efficiency (cf. Figure 10).

Figures 11 and 14 show typical fiber-matrix bond condition for various fiber in gypsumpulp and gypsum-sankelite matrix. Also, the matrix-aggregate condition of these two type of matrices are shown in Figure 11. The following observations show:
(a) For the gypsum matrix, the interfacial region appears to be highly porous compared to the more solid and dense structure of cement matrix. At higher magnification, elongated needle like gypsum crystals ( $\mathrm{CaSO}_{4} \cdot \mathrm{nH}_{2} \mathrm{O}$ ) are clearly identifiable in Figure 14. Capillary voids can be seen to be many times larger than cement matrix, and this observed characteristics were found to be common for both gypsum-pulp and gypsumsankelite.
(b) It can be seen in the micrographs in Figure 11-14 that bonding mechanism is essentially a mechanical interlocking and physicochemical bonding. The rough surface and multistrand structure of fibers offered extra anchoring points to the matrix especially in the case of sisal and polypropylene fiber hence giving an excellent mechanical bonding strength property. Crystal growth deposited at the surface interstices of the fibers are also revealed physicochemical bonding. It is a point of interest to see that crystal deposits at the polypropylene fiber surface are minimal compared to the PVA vinylon and natural fibers (Figure 12). This confirms the hydrophobic surface quality of the polypropylene fiber. However for the same polymer based vinylon fiber, it was observed that strong bond of matrix to the fiber existed, including that its physicochemical bonding is far excellent than the polypropylene. Akers (1989) noted that PVA fibers have good affinity with water and cement matrix because of its polymer structure, that is, the possession of hydroxyl group essential for bonding with waterbased binder such as cement or gypsum. In the case of natural fibers, as expected, the excellent surface wettability of natural fibers provided an increased physicochemical bonding with gypsum matrix (Figure 13). It can be seen that crystal deposits presumably gypsum gel and $\mathrm{CaSO}_{4} \cdot n \mathrm{H}_{2} \mathrm{O}$ crystals accumulated at the fiber surface and interstices.
(c) Gypsum crystal formed at the interfacial region seemed to provide a good interlocking mechanism. However, the aggregate-matrix interfacial bond appears porous and, a closer look at the sankelite sand surface would show that there is no clear evidence of physicochemical bonding between the rich $\mathrm{SiO}_{2}$ sankelite sand and the gypsum hydration crystals.

## Effect of W/G ratio and lightweight aggregate (pulp and sankelite) to the Fresh and Mechanical properties of plain gypsum mortar

Table 5 lists the fresh and mechanical properties of plain gypsum mortar. Figures 15-19 show the various parameters which affects the fresh and mechanical properties of a gypsum pulp or gypsum-sankelite mortar. Findings show:
(a) Significant decrease in the workability in terms of flow value of the gypsum paste with an increasing cellulose pulp or sankelite sand content was observed (Figure 15). This decrease is due to the filler effects of aggregates and the high water absorption of pulp and sankelite aggregates. However this property can be greatly improved through the introduction of superplastisicer. For this two types of lightweight aggregate the cellulose pulp showed the greatest water absorption, which can later pose some problems with workability and effective fiber volume addition. The angular and porous structure as observed from the SEM micrograph are believed to be the major factors causing this behavior (cf. Figures 11-13).
(b) It is generally known that $\alpha$-gypsum sets very fast leaving no ample time for proper casting and vibration of the mixture in forms. Hence, an inorganic retarder was used to delay the setting time, and various percentage dosage to gypsum matrix were examined. Figure 16 shows the effect of retarder to setting time. Based from this results, an
optimum dosage of $1 \%$ is most recommendable, such that a delay in the final setting time to about 1.25 hours can be achieved.
(c) The effect of W/G ratio to both bending and compressive strength shows that it follows the typical trend found in cement based composite, that is, at an increasing C/W ratio a linearly increasing ( Fc ') strength occurs. It can be seen in Figure 17 that variation in the linear relationship is affected by the aggregate proportion, especially in the case of cellulose pulp where inherent high water absorption property lead to reduce workability and overall change in the effective W/G ratio of the mixture.
(d) The maximum proportion of lightweight pulp or Shirasu aggregate that will still produce a highly workable mix were found to be in the range of $10-20 \%$. Since the objective of the study is to fabricate lightweight, low-cost but of sufficient mechanical strength building element, it is necessary to adopt a high aggregate proportion of 10 and $15 \%$ were found to be the most appropriate proportions based on the criteria of workability and strength. Figure 18 shows the relationship of gypsum-mortar unit weight to its strength properties. Note that for a typical unit weight of $1400 \mathrm{~kg} / \mathrm{m}^{3}$ a compressive and bending strength of about 200 and $60 \mathrm{kgf} / \mathrm{cm}^{2}$, respectively, can be attained at various proportions.
(e) The flexural modulus of elasticity as computed from the bending stress-strain curves shows that for a typical compressive strength of $150-250 \mathrm{kgf} / \mathrm{cm}^{2}$ and unit weight of $1600 \mathrm{kgf} / \mathrm{m}^{3}$, the elastic modulus ranged from 150000 to $200000 \mathrm{kgf} / \mathrm{cm}^{2}$ (Figure 19). Compared with the empirical equation of the elastic modulus given by the AIJ (Architectural Institute of Japan, 1993) or ACI (American Concrete Institute, 1993), the gypsum mortar likewise follows the same trend as the typical concrete.
(f) Figure 20 shows the bending and compressive strength development of various gypsum-pozzolan-pulp mortar at standard curing condition. It can be observed that increase in strength by as much as $20 \%$ occurred as early as 7 days, and tapered off to $11 \%$ at 63 days. This behavior can be seen for all pozzolanic material especially at $10 \%$ gypsum replacement. The order of strength increase contribution of various pozzolanic materials was seen as (1) silica fume (2) white silica and (3) fly-ash. Contrary to expectation, the $10 \%$ substitution by mineral admixture showed marginal strength increase which the author presume to be caused by increase viscosity in the gypsum mixture due to the absence of superplastisicer. Otherwise it might produce a more appreciable strength increase which is commonly evident for matrices with pozzolanic additions.

## Effect of fiber reinforcement to gypsum-pulp and gypsum-sankelite mortar

Table 6 list the fresh and mechanical properties of gypsum fiber composite. Figure 21 to 23 show the effect of various parameter to the flexural and deformation properties of various gypsum fiber concrete. From these results, it can be observed that:
(a) Due to the lesser workability of the gypsum-pulp matrix caused by the higher absorption index of cellulose pulp, the introduction of fiber was limited to a maximum of $2 \%$, that
is in the absence of high range water reducing admixture (HRWR) or superplastisicer. While, for gypsum-sankelite a fiber introduction of $3 \%$ was attained with still sufficient workability property. The difference in the workability of the two matrices is due to the rheological relationship of aggregates and fiber, and this can be confirmed from SEM micrographs.
(b) It can be seen in Table 6 that there is no significant increase in the compressive strength of the composite with an increase in fiber such as polypropylene and the natural fibers especially at higher fiber volume. This trend is generally expected for low-modulus types of fibers because such fibers do not contribute any reinforcement against compression. Theoretically, this behavior can be proven by the simple law of mixture, that is

$$
\begin{equation*}
E_{c}=E_{m}\left(1-V_{f}\right)+\eta_{o}, \eta_{l}, \eta_{b} E_{f} V_{f} \tag{1}
\end{equation*}
$$

where $E, V, \sigma, \epsilon$ refer to elasticity, volume fraction, general stress and strain of constituent material, and subscripts $c, m$ and $f$ refer to composite, matrix and fiber respectively, $n o, n l, n b$ are efficiency factors for fiber orientation, length and bonding condition, respectively. The equation refers to the contribution to strength of matrix and fiber components of the fiber composite.

If Equation 1 is normalized by the elastic modulus of matrix ( Em ),

$$
\begin{aligned}
& \frac{E_{c}}{E_{m}}=\left(1-V_{f}^{\prime}\right)+\eta_{o}, \eta_{l}, \eta_{b} \frac{E_{f}}{E_{m}} V_{f} \\
& \frac{E_{c}}{E_{m}}=1+V_{f}\left(\eta_{o}, \eta_{l}, \eta_{b} \frac{E_{f}}{E_{m}}-1\right)
\end{aligned}
$$

Assuming that the strain developed by the fiber and the matrix at the interface is equal ( $\epsilon_{\mathrm{f}}$ $=\epsilon_{\mathrm{m}}$ ), equation 5 becomes

$$
\begin{align*}
& \frac{\sigma_{c}}{\sigma_{m}}=1+V_{f}\left(\eta_{o}, \eta_{l}, \eta_{b} \frac{E_{f}}{E_{m}}-1\right) \\
& \sigma_{c}=\sigma_{m}\left(1+V_{f}\left(\eta_{o}, \eta_{l}, \eta_{b} \frac{E_{f}}{E_{m}}-1\right)\right) \tag{2}
\end{align*}
$$

$$
\begin{equation*}
\sigma_{c}=\sigma_{m} \cdot \psi \tag{3}
\end{equation*}
$$

Hence if the elastic modulus of the fiber ( $E_{f}$ ) is far less than the elastic modulus of matrix $\left(E_{m}\right)$ and coupled with a relatively large fiber volume fraction $\left(V_{f}\right)$, then the factor $\psi$ would be less than one, thereby resulting to a compressive strength ( $\sigma_{c}$ ) less than that of the plain unreinforced matrix. This simple derivation confirm the trend observed in Table 6. Furthermore as a confirmation, in the case of vinylon and glass fiber having an elastic modulus of about 30 and 70 GPa which is about $1.25-3$ times that of the plain matrix, test results showed a slight improvement in the compressive strength.
(c) For both gypsum-pulp and gypsum-Shirasu matrix, increase in first crack strength at 7 days age due to reinforcement of vinylon, polypropylene, sisal and coconut fiber is quite significant, that is, at the range of $15-25 \%$. This characteristic may be due to relatively. low elastic modulus ( $\mathrm{Ec} \approx 130,000 \mathrm{~kg} / \mathrm{cm}^{2}$ ), flexural strength ( $\sigma_{\mathrm{c}} \approx 50 \mathrm{kgf} / \mathrm{cm}^{2}$ ) and ductile deformation property of the gypsum matrix, hence giving the fiber adequate chance to transfer and redistribute stresses throughout the matrix system. This characteristic is different to that observed in cement matrix where increase in first crack strength even with high modulus fiber like steel and glass where only limited to about $15 \%$ (ACI-544, 1989; Swamy, 1990).
(d) For fiber volume of $1 \%$ improvement to ultimate bending strength is fairly negligible as can be seen in the load-deformation diagram for various fiber reinforcement and for both gypsum-pulp and gypsum-sankelite matrix (cf. Figures 21 and 22). However as fiber volume expected, the relatively strong PVA-vinylon and sisal fiber showed an excellent post-crack characteristics, indicating a high energy absorption capability. This may be attributed to their excellent fiber-matrix bond condition as shown by the SEM micrograph (cf. Figures 5-8). The polypropylene and coconut fiber, on the other hand, showed only a fair ductile characteristics. This observed lower post-crack property is presumably due to lesser fiber-matrix bond strength as seen from the relatively smooth surface of coconut fibers and to the hydrophobic surface of polypropylene fiber. Hence the difference in the post-crack strength and toughness of a fiber composite can be greatly attributed to its varying fiber-matrix bonding mechanism.
(e) Since one of the aim for this developmental study is on the investigation of the feasibility of gypsum-natural fiber mortar for building elements, various panel sections were investigated. Figure 21-23 show the variation of strength according to shear-span ratio or specimen dimension. It can be seen that the ultimate bending strength of the panel decreases with the decrease in the shear-span ratio or volume of stressed material. In spite of this, the load-deformation behavior is fairly consistent in all specimen dimension especially in terms of toughness and ductility of the gypsum-fiber composite material. It was also observed that for efficient fiber reinforcement such as sisal, the effect of fiber volume to strength under various specimen scale dimension is very significant, that is, for thinner sections the ductility and strength contribution of efficient and strong fibers are more significant compared to other efficient fibers such as coconut.

## CONCLUSION

This comparative and developmental research study has shown that the utilization of natural fibers such as coconut and sisal to a gypsum matrix is highly feasible. Utilization of natural fibers can solve two significant problems, namely, elimination of solid waste and the provision of a valuable construction materials. Based from the experimental results, the following significant conclusion can be drawn:

1. SEM micrograph revealed insights on the structure of various pozzolan and various lightweight aggregate materials. Morphological characteristics that influences workability, strength development and bonding to cement matrix of pozzolan and lightweight material were clearly observed. Morphological and physicochemical interlocking between gypsum hydrated products and fibers govern both the bonding mechanism, crack deflection and arrest mechanism.
2. Increasing the lightweight aggregate volume can considerably decrease the flowability of the gypsum matrix. Hence, a highly flowable but relatively strong gypsum matrix with $\mathrm{W} / \mathrm{G}=35-45 \%$ and $\mathrm{S} / \mathrm{G}$ ratio of $10-20 \%$ was adopted as the most appropriate mix, which at the same time can enable a $2 \%$ fiber volume reinforcement to its matrix.
3. The gypsum matrix with either pulp or sankelite follows the same linear C/W-strength relationship as the cement matrix. A typical compressive strength of $150-200 \mathrm{kgf} / \mathrm{cm}^{2}$ and bending strength of $50-60 \mathrm{kgf} / \mathrm{cm}^{2}$ can have an average unit weight of $1500 \mathrm{kgf} / \mathrm{cm}^{2}$ and an elastic modulus of $150000-200000 \mathrm{kgf} / \mathrm{cm}^{2}$.
4. Mineral admixture can increase the strength by $20 \%$ at 7 days and $11 \%$ at 63 days, and continuously to increase with time. Apparent optimum gypsum substitution by weight in this experiment showed to be $10 \%$. Ability of either pozzolanic material or lightweight aggregate to improve workability are highly dependent on its physical and morphological structure.
5. Fiber reinforcement to gypsum matrix showed positive results compared to cement based matrix. Increase in 7 -day strength by $15-25 \%$ due to either vinylon, polypropylene, sisal and coconut fiber reinforcement occurred. The principle of Law of Mixture also satisfactorily describes the contribution of both the fiber and matrix to every mechanical properties of gypsum fiber composite.
6. In terms of ductility and toughness, gypsum matrices reinforced with vinylon and sisal showed an excellent load-deformation behavior, followed by the polypropylene and coconut fiber. The observed difference in ductility are attributed to the different degree of bonding between fiber and matrix.
7. Scale effect is very significant for fiber concrete in various panel dimension. Strength efficient fiber such as sisal magnifies the scale effect at various fiber volume fraction, while for lesser efficient fiber such as sisal magnifies the scale effect at various fiber volume fraction, while for lesser efficient fiber such as coconut this effect appears to be negligible. Generally from the test results, the fiber volume fraction followed by the gypsum matrix proportions are the two major factors that influence the scale effect in panels.

## ACKNOWLEDGEMENT

The author wishes to express his sincerest thanks to Mr. Yasuo Maruyama of the Institute of Technology, Shimizu Corporation for his assistance in the experiments. Special thanks is also due to the rest of the staff of the Civil Engineering Department, Integrated Research and Training Center, Technological University of the Philippines for their comments and suggestion during the experimental stage of this project.

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Table 5
Fresh and Mechanical Properties Plain Gypsum Mortar at Various W/G and S/G Ratio

| MLXCODE | FIBER |  | MIX PROPORTION |  |  | FRESH PROPERTIES |  | BENDING STRENGTH |  |  |  | COMPRESSIVE STRENGTH $\mathrm{kg} / \mathrm{cm}^{2}$ |  |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | TYPE | VF | $\begin{aligned} & \text { W/G } \\ & \text { (\%) } \end{aligned}$ | S/G <br> (\%) | POZZ F/F +C <br> (\%) | UNIT WT (g/cc) | $\begin{aligned} & \text { FLOW } \\ & (\mathrm{mm}) \end{aligned}$ | FIRST CRACK |  | POST CRACK |  | 7-DAY | $\begin{gathered} \hline 28- \\ \text { DAY } \end{gathered}$ |
|  |  | (\%) |  |  |  |  |  | 7-DAY | 28-DAY | 7-DAY | $\begin{gathered} 28- \\ \text { DAY } \end{gathered}$ |  |  |
| G35 | PLAIN GYPSUM | 0 | 35 | 10 | 0 | 1.85 | 248 | 69.5 | 78.3 | 0.0 | 0.0 | 208 | 244 |
| SERIES |  | 0 | 35 | 15 | 0 | 1.85 | 176 | 70.6 | 68.0 | 0.0 | 0.0 | 230 | 250 |
|  |  | 0 | 35 | 20 | 0 | 1.89 | 138 | 70.0 | 79.5 | 0.0 | 0.0 | 207 | 282 |
| G45 | PLAIN GYPSUM | 0 | 45 | 10 | 0 | 1.83 | flowing | 63.5 | 89.1 | 0.0 | 0.0 | 194 | 226 |
| SERIES |  | 0 | 45 | 15 | 0 | 1.82 | 235 | 58.3 | 68.1 | 0.0 | 0.0 | 183 | 206 |
|  |  | 0 | 45 | 20 | 0 | 1.81 | 155 | 63.6 | 61.2 | 0.0 | 0.0 | 201 | 217 |
| GS35 | PLAIN GYPSUM | 0 | 35 | 10 | 0 | 1.42 | 173 | 45.1 | 51.0 | 0.0 | 0.0 | 195 | 228 |
| SERIES |  | 0 | 35 | 15 | 0 | 1.39 | 143 | 40.0 | 55.2 | 0.0 | 0.0 | 180 | 213 |
| GS45 | PLAIN GYPSUM | 0 | 45 | 10 | 0 | 1.34 | flowing | 38.8 | 49.6 | 0.0 | 0.0 | 150 | 195 |
| SERIES |  | 0 | 45 | 15 | 0 | 1.30 | 220 | 36.3 | 52.3 | 0.0 | 0.0 | 159 | 198 |
|  |  | 0 | 45 | 20 | 0 | 1.22 | 160 | 29.3 | 51.6 | 0.0 | 0.0 | 119 | 190 |

Table 6
Fresh and Mechanical Properties of Gypsum Fiber
Mortar at 7-Days

| MLX- <br> CODE | FIBER | MLX PROPORTION |  |  |  | FRRESH PROPERTY |  | BENDING STRENGTH ( $\mathrm{kg} / \mathrm{cm} 2$ ) |  | COMPRES <br> SIVE <br> STRENGTH |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
|  | TYPE | VF <br> (\%) | W/G <br> (\%) | S/G <br> (\%) | POZZ <br> F/F+C <br> (\%) | UNIT WT $(\mathrm{g} / \mathrm{cc})$ | FLOW <br> (mm) | FIRST <br> CRACK | POST CRACK | (kgf/cm2) |
| V1-G45 | VINYLON | 1 | 45 | 15 | 0 | 1.548 | 203 | 71.8 | 70.0 | 231 |
| V2-G45 |  | 2 | 45 | 15 | 0 | 1.605 | 185 | 720 | 98.0 | 252 |
| V1-G35 |  | 1 | 35 | 10 | 0 | 1.585 | 168 | 76.6 | 42.3 | 265 |
| P1-G45 | POLYPRO <br> PYLENE | 1 | 45 | 15 | 0 | 1.660 | 172 | 63.7 | 37.0 |  |
| P2-G45 |  | 2 | 45 | 15 | 0 | 1.548 | 140 | 63.1 | 53.5 | 237 |
| P1-G35 | POLYPRO <br> PYLENE | 1 | 35 | 10 | 0 | 1.617 | 148 | 70.4 | 45.3 | 274 |
| S1-G45 | SISAL | 1 | 45 | 15 | 0 | 1.603 | 183 | 67.4 | 43.0 | 247 |
| S2-G45 |  | 2 | 45 | 15 | 0 | 1.600 | 147 | 68.8 | 67.0 | 232 |
| S1-G35 | SISAL | 1 | 35 | 10 | 0 | 1.585 | 150 | 74.5 | 43.4 | 217 |
|  | COCONUT | 1 | 45 | 15 | 0 | 1.470 | 180 | 69.1 | 22.0 | 223 |
| C2-G45 |  | 2 | 45 | 15 | 0 | 1.550 | 152 | 76.8 | 38.0 | 209 |
| C2-GS35 | COCONUT | 2 | 35 | 10 | 0 | 1.457 | 148 | 52.4 | 36.2 | 218 |
| C2-GS45 |  | 2 | 45 | 15 | 0 | 1.301 | 163 | 36.1 | 27.7 | 142 |
| C3-GS45 |  | 3 | 45 | 10 | 0 | 1.245 | 153 | 36.3 | 37.0 | 183 |
| S2-GS35 | SISAL | 2 | 35 | 10 | 0 | 1.505 | 145 | 58.1 | 55.4 | 219 |
| S2-GS45 |  | 2 | 45 | 15 | 0 | 1.303 | 158 | 69.5 | 72.0 | 156 |
| S3-GS45 |  | 3 | 45 | 10 | 0 | 1.303 | 146 | 72.2 | 100.0 | 161 |
| P2-GS35 | POLYPRO PYLENE | 2 | 35 | 10 | 0 | 1.512 | 145 | 54.7 | 48.2 | 217 |
| P2-GS45 |  | 2 | 45 | 15 | 0 | 1.321 | 151 | 32.4 | 36.0 | 136 |



Figure 2(a) X-ray Flouresence Spectral Analysis of Silica Fume


Figure 2(b) X-ray Flouresence Spectral Analysis of White Silica


Fig.3. SEM micrograph of Fly-ash showing the smooth-round morphology


Fig.5. Hollow and smooth structure Shirasu balloon (sankelite sand)


Fig.7. Surface and section morphology of Coconut fibers


Fig.4. Morphology of White silica showing the angular and porous structure


Fig.6. Angular and porous structure Cellulose (paper)pulp


Fig.8. Structure of surface and section of Sisal fibers


Fig.9. Morphology of vinylon fiber, showing the close and dense surface


Fig.11. Sankelite aggt-matrix interfacial structure, showing the porous gypsum


Fig.13. Coconut fiber-matrix morphology


Fig.10. Morphology of polypropylene, showing the synthetic multi-strand structure


Fig.12. Vinlyon-gypsum matrix interfacial structure


Fig.14. Magnified gypsum crystals at interfacial region of fiber and matrix


Fig.15. Trend of flow values for gypsum mortar without superplastisicer at various pulp content


Fig.16. Effect of retarder dosage to final setting of plain gypsum-pulp mortar


Fig. 17. Effect of W/G ratio to the compressive strength of plain gypsum mortar


Fig.18. Relationship of unit weight and bending strength at 7 days for gypsum mortar with W/G $=35 \%$


Fig.19. Modulus of Elasticity and strength relationship of gypsum mortar at 7 days



Fig.21(a) Stress-deflection of gypsum based mortar at 7 days with $W / G=45 \%$, pulp $=20 \%$ and $V F=1 \%$ for $4 \times 4 \times 16$


Fig.21(b) Stress-deflection of gypsum based mortar at 7 days with $W / G=40 \%$, pulp $=20 \%$ and $V F=2 \%$ for $4 \times 4 \times 16$


Fig.22(a) Stress-deformation of gypsum fiber composite $60 \times 30 \times 2$ board with Vf $=2 \%, W / G=45 \%$


Fig.22(b) Stress-deformation of gypsum sankelite fiber composite $60 \times 30 \times 2$ board with $\mathrm{Vf}=3 \%$, $\mathrm{W} / \mathrm{G}=45 \%$


Shear-span ratio (A/D)
Specimen dimension (mm)

Fig. 23 Effect of specimen dimension to the bending strength of gypsum-fiber mortar


[^0]:    * limitations indicate where a specific variable was applied

